

# Ba<sub>2</sub>An(S<sub>2</sub>)<sub>2</sub>S<sub>2</sub> (An = U, Th): Syntheses, Structures, Optical, and Electronic Properties

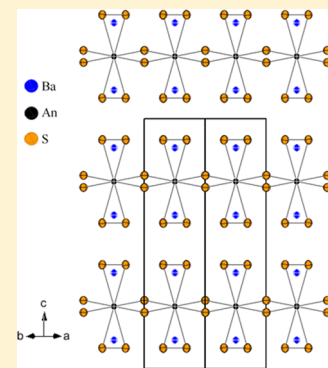
Adel Mesbah,<sup>†</sup> Emilie Ringe,<sup>†</sup> Sébastien Lebègue,<sup>‡</sup> Richard P. Van Duyne,<sup>†</sup> and James A. Ibers<sup>\*,†</sup>

<sup>†</sup>Department of Chemistry, Northwestern University, 2145 Sheridan Road, Evanston, Illinois 60208-3113, United States

<sup>‡</sup>Laboratoire de Cristallographie, Résonance Magnétique, et Modélisations CRM2 (UMR UHP-CNRS 7036), Faculté des Sciences et Techniques, Université de Lorraine, BP 70239, Boulevard des Aiguillettes, 54506 Vandoeuvre-lès-Nancy Cedex, France

## Supporting Information

**ABSTRACT:** The compounds Ba<sub>2</sub>An(S<sub>2</sub>)<sub>2</sub>S<sub>2</sub> (An = U, Th) have been synthesized by reactions of the elements with BaS and S at 1273 and 1173 K, respectively. These isostructural compounds crystallize in a new structure type in the tetragonal space group  $D_{4h}^{15}-P4_2/nmc$ . The structure comprises Ba<sup>2+</sup> cations and  ${}_{\infty}^2[\text{An}(\text{S}_2)_2(\text{S})_2^{4-}]$  layers. The An<sup>4+</sup> cations in these layers are arranged linearly and are bridged by S<sup>2-</sup> anions. Coordination about the An center, which has symmetry  $\bar{4}m2$ , consists of two S<sub>2</sub><sup>2-</sup> ions and four S<sup>2-</sup> ions. Thus, the compounds are charge-balanced with An<sup>4+</sup>. No other alkali-metal actinide chalcogenides are known that contain chalcogen–chalcogen bonds. Optical measurements on Ba<sub>2</sub>Th(S<sub>2</sub>)<sub>2</sub>S<sub>2</sub> indicate a direct band gap of 2.46(5) eV. Density functional theory calculations, performed with the HSE exchange–correlation potential, lead to band gaps of 2.2 and 1.8 eV for Ba<sub>2</sub>Th(S<sub>2</sub>)<sub>2</sub>S<sub>2</sub> and Ba<sub>2</sub>U(S<sub>2</sub>)<sub>2</sub>S<sub>2</sub>, respectively, thus demonstrating the utility of applying this functional to 5f-electron systems.



## INTRODUCTION

The crystal chemistry of the actinide chalcogenides An/Q (An = 5f element; Q = S, Se, or Te) has been widely explored. The presence of the 5f elements results in compounds that display varied structural, electronic, magnetic, and optical properties.<sup>1,2</sup> These compounds are usually obtained by solid-state syntheses involving the reactive flux method<sup>3</sup> or by direct combination of the elements.<sup>2,4</sup>

Relatively few binary, ternary, and quaternary actinide chalcogenides exhibit Q–Q bonding (Q = S, Se, or Te). Among the sulfide binaries are US<sub>3</sub><sup>5</sup> and An<sub>2</sub>S<sub>5</sub> (An = Th, U).<sup>6–9</sup> Among the ternaries that show Q–Q interactions are CsUTe<sub>6</sub>,<sup>10</sup> K<sub>4</sub>USE<sub>8</sub>,<sup>11</sup> and the relatively large family of AAAn<sub>2</sub>Q<sub>6</sub> compounds (with A = K, Rb, or Cs; An = U, Th, or Np, Q = S, Se, or Te) whose structure types are CsTh<sub>2</sub>Te<sub>6</sub><sup>12</sup> and KTh<sub>2</sub>Se<sub>6</sub>.<sup>13</sup> The quaternaries RbSbU<sub>2</sub>S<sub>8</sub><sup>14</sup> and Ta<sub>2</sub>UO(S<sub>2</sub>)<sub>3</sub>Cl<sub>6</sub><sup>15</sup> display S–S bonds.

All but the last compound are binaries or are formed by the insertion of a monovalent cation to form a ternary or a quaternary. The insertion of an alkali-metal cation (Ak) to form a ternary or a quaternary (with further insertion of another cation) remains largely uninvestigated. In the Ak/An/Q family, the compounds BaUS<sub>3</sub>,<sup>16,17</sup> AkU<sub>2</sub>S<sub>5</sub> (Ak = Ca, Sr, Ba),<sup>18,19</sup> and SrTh<sub>2</sub>Se<sub>5</sub><sup>4</sup> are known. Several quaternaries (Ak/An/M/Q) are also known. These include Ba<sub>2</sub>Cu<sub>2</sub>US<sub>5</sub>,<sup>20</sup> Ba<sub>4</sub>Cr<sub>2</sub>US<sub>9</sub>,<sup>20</sup> and Ba<sub>8</sub>Hg<sub>3</sub>U<sub>3</sub>S<sub>18</sub>.<sup>21</sup> None of these contain Q–Q bonds.

One objective of the present work was to explore the synthesis of actinide chalcogenides containing an alkaline-earth metal. The size and charge of an alkaline-earth metal, as opposed to those of an alkali metal, will surely lead to new compounds and structures. In the present work, the syntheses

and structure of the two new isostructural compounds Ba<sub>2</sub>U(S<sub>2</sub>)<sub>2</sub>S<sub>2</sub> and Ba<sub>2</sub>Th(S<sub>2</sub>)<sub>2</sub>S<sub>2</sub> are described. These differ from the other Ak/An/Q compounds in that they contain Q–Q single bonds. A second objective of the present study was to assess the utility of the HSE functional in density functional theory (DFT) calculations of band gaps in 5f-electron systems. We thus describe some optical properties of Ba<sub>2</sub>Th(S<sub>2</sub>)<sub>2</sub>S<sub>2</sub> and the electronic properties of both compounds as calculated by the DFT method performed with the HSE exchange–correlation potential.

## EXPERIMENTAL METHODS

**Syntheses.** The following reactants were used as starting materials: <sup>238</sup>U powder, obtained by hydridization and decomposition of U turnings (ORNL) in a modification<sup>22</sup> of a literature method,<sup>23</sup> Th powder (MP Biomedicals, LLC 99.1%), BaS (Alfa, 99.7%), and S (Mallinckrodt, 99.6%). The reactions were performed in sealed carbon-coated fused-silica tubes. The starting mixtures were loaded into such tubes under an Ar atmosphere in a glovebox. Then the tubes were evacuated to 10<sup>−4</sup> Torr, flame-sealed, and placed in a computer-controlled furnace. The elemental composition was determined qualitatively with an energy-dispersive X-ray (EDX)-equipped Hitachi S-3400 scanning electron microscope.

**Ba<sub>2</sub>U(S<sub>2</sub>)<sub>2</sub>S<sub>2</sub>.** Very small black blocks of Ba<sub>2</sub>U(S<sub>2</sub>)<sub>2</sub>S<sub>2</sub> in about 20 wt % yield based on U were obtained by the reaction of BaS (21.6 mg, 0.128 mmol), U (20.2 mg, 0.085 mmol), and S (9.5 mg, 0.298 mmol). The reaction mixture was heated to 1273 K in 48 h, held there for 4 h, cooled to 1223 K in 12 h, kept there for 8 days, and then cooled to 298 K at 3 K/h. EDX analysis performed on selected black blocks showed

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Ba/U/S in the approximate ratio 2:1:6. The major product of the reaction was polycrystalline UOS.

**Ba<sub>2</sub>Th(S<sub>2</sub>)<sub>2</sub>S<sub>2</sub>.** This compound was synthesized from the mixture of BaS (28.8 mg, 0.17 mmol), Th (19.8 mg, 0.084 mmol), and S (16.35 mg, 0.51 mmol). This mixture was heated to 1173 K, kept there for 4 days, and then cooled to 473 K at 3 K/h, and then the furnace was turned off. Green plates in about 50 wt % yield based on Th were isolated and analyzed by EDX. These showed Ba/Th/S in the approximate ratio 2:1:6. The other product was ThOS, as judged by its color and the habit of the crystals, together with EDX analysis of Th:S = 1:1.

**Structure Determinations.** Single-crystal X-ray diffraction data for Ba<sub>2</sub>U(S<sub>2</sub>)<sub>2</sub>S<sub>2</sub> and Ba<sub>2</sub>Th(S<sub>2</sub>)<sub>2</sub>S<sub>2</sub> were collected with the use of graphite-monochromatized Mo K $\alpha$  radiation ( $\lambda = 0.71073$  Å) at 100 K on a Bruker APEX2 diffractometer.<sup>24</sup> The data collection strategy was optimized with the algorithm COSMO in the program APEX2<sup>24</sup> as a series of 0.3° scans in  $\omega$  and  $\phi$ . The exposure time was 10 s/frame. The crystal-to-detector distance was 6 cm. The collection of the intensity data as well as cell refinement and data reduction was carried out with the use of the program APEX2.<sup>24</sup> Face-indexed absorption, incident beam, and decay corrections were performed with the use of the program SADABS.<sup>25</sup> Both structures were solved and refined with the use of the SHELXTL programs.<sup>26</sup> The program STRUCTURE TIDY<sup>27</sup> in PLATON<sup>28</sup> was used to standardize the atomic positions. Further details are given in Table 1 and in the Supporting Information.

**Table 1. Crystal Data and Structure Refinements for Ba<sub>2</sub>U(S<sub>2</sub>)<sub>2</sub>S<sub>2</sub> and Ba<sub>2</sub>Th(S<sub>2</sub>)<sub>2</sub>S<sub>2</sub><sup>a</sup>**

|   | Ba <sub>2</sub> U(S <sub>2</sub> ) <sub>2</sub> S <sub>2</sub> | Ba <sub>2</sub> Th(S <sub>2</sub> ) <sub>2</sub> S <sub>2</sub> |
|---|--|---|
| fw (g mol <sup>-1</sup> )   | 705.07   | 699.08  |
| color   | black  | green   |
| <i>a</i> (Å)  | 5.4145(2)  | 5.4810(2)   |
| <i>c</i> (Å)  | 15.7322(5)   | 15.9860(6)  |
| <i>V</i> (Å <sup>3</sup> )  | 461.22(3)  | 480.24(4)   |
| $\rho$ (g cm <sup>-3</sup> )  | 5.077  | 4.834   |
| $\mu$ (mm <sup>-1</sup> )   | 27.229   | 24.776  |
| <i>R</i> ( <i>F</i> ) <sup>b</sup>  | 0.0144   | 0.0187  |
| <i>R</i> <sub>w</sub> ( <i>F</i> <sub>o</sub> <sup>2</sup> ) <sup>c</sup> | 0.0346   | 0.0378  |

<sup>a</sup>For both structures, space group  $D_{4h}^{15}P4_2/nmc$ ,  $Z = 2$ ,  $\lambda = 0.71073$  Å, and  $T = 100(2)$  K. <sup>b</sup> $R(F) = \sum ||F_o| - |F_c|| / \sum |F_o|$  for  $F_o^2 > 2\sigma(F_o^2)$ . <sup>c</sup> $R_w(F_o^2) = \{ \sum [w(F_o^2 - F_c^2)]^2 / \sum w F_o^4 \}^{1/2}$ . For  $F_o^2 < 0$ ,  $w^{-1} = \sigma^2(F_o^2)$ ; for  $F_o^2 \geq 0$ ,  $w^{-1} = \sigma^2(F_o^2) + (qF_o^2)^2$ , where  $q = 0.0106$  for Ba<sub>2</sub>U(S<sub>2</sub>)<sub>2</sub>S<sub>2</sub> and 0.0183 for Ba<sub>2</sub>Th(S<sub>2</sub>)<sub>2</sub>S<sub>2</sub>.

**Optical Measurements.** Optical absorption measurements for Ba<sub>2</sub>Th(S<sub>2</sub>)<sub>2</sub>S<sub>2</sub> were performed over the range from 3.76 eV (330 nm) to 1.39 eV (894 nm) at 293 K on four and six different regions along the (001) crystal faces of two distinct single crystals. Each was mounted on a goniometer head and inserted on a custom-made holder fitted to a Nikon Eclipse Ti2000-U inverted microscope.<sup>29,30</sup> The crystal was positioned at the focal plane above the 20 $\times$  objective of the microscope and illuminated with a tungsten–halogen lamp. The transmitted light was spatially filtered with a 200  $\mu$ m aperture, dispersed by a 150 grooves/mm grating in an Acton SP2300i imaging spectrometer, and collected on a back-illuminated, LN<sub>2</sub>-cooled charge-coupled device (Spec10:400BR, Princeton Instruments). Polarization experiments were performed by inserting a polarizer between the lamp and crystal; spectra were taken at 10° increments of the polarizer angle.

Similar measurements on Ba<sub>2</sub>U(S<sub>2</sub>)<sub>2</sub>S<sub>2</sub> were not possible because of the small size of the crystals and contamination of Ba<sub>2</sub>U(S<sub>2</sub>)<sub>2</sub>S<sub>2</sub> powder by UOS.<sup>31,32</sup>

**Computational Calculations.** The calculations were performed within the framework of DFT<sup>33</sup> together with the exchange–correlation potential HSE06.<sup>34–36</sup> In this formulation, the exchange potential contains a part of the Hartree–Fock potential for the short-range part of the interaction, while the correlation potential is kept at the

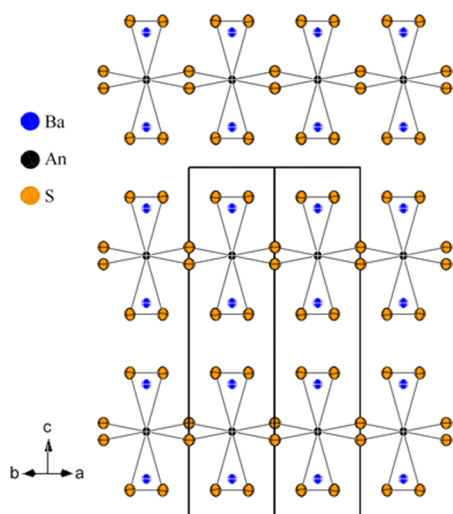
generalized gradient approximation (GGA) level.<sup>37</sup> In particular, HSE06 is known to cure partly the problem of self-interaction and also to provide band gaps that usually are in reasonable agreement with the experiment.<sup>38</sup> Although the HSE functional is becoming increasingly popular, it has not been applied widely to systems containing *f* electrons, although one can cite, for example, works on UO<sub>2</sub>, PuO<sub>2</sub>, and Pu<sub>2</sub>O<sub>3</sub><sup>39</sup> and on CeO<sub>2</sub> and Ce<sub>2</sub>O<sub>3</sub>,<sup>40–42</sup> with satisfactory results in comparison to the available experiments *The Vienna ab Initio Simulation Package* (VASP)<sup>43,44</sup> implementing the projector-augmented wave method<sup>45</sup> was used to perform the ab initio simulations. A 6  $\times$  6  $\times$  2 *k*-point mesh was used to sample the Brillouin zone together with the default cutoff for the plane-wave expansion. The lattice parameters and positions of the atoms were kept identical with those obtained experimentally. Spin polarization was allowed, and the two possible configurations (ferromagnetic or antiferromagnetic) within the crystal cell were checked. Also, the optical properties were calculated using the HSE eigenvalues and orbitals in the sum-over-states approximation, as implemented<sup>46</sup> in the VASP code.

## DISCUSSION

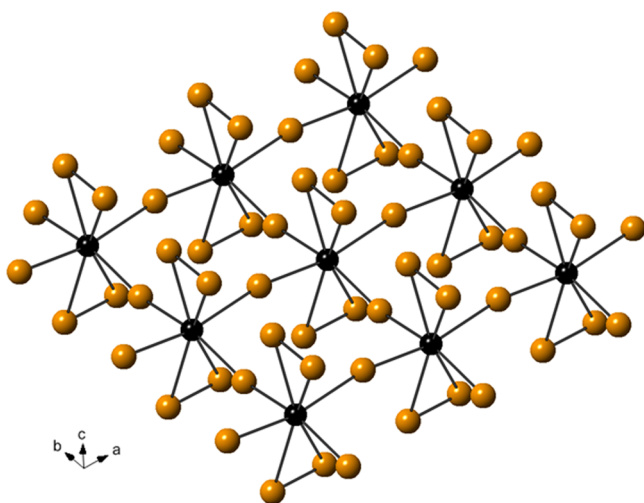
**Syntheses.** The compounds Ba<sub>2</sub>U(S<sub>2</sub>)<sub>2</sub>S<sub>2</sub> and Ba<sub>2</sub>Th(S<sub>2</sub>)<sub>2</sub>S<sub>2</sub> were synthesized by the reactions of U or Th with BaS and S at 1273 and 1173 K, respectively. The yield of very small black blocks of Ba<sub>2</sub>U(S<sub>2</sub>)<sub>2</sub>S<sub>2</sub> was about 20 wt % based on U, with the other product being UOS. No single crystals larger than about 40  $\mu$ m could be made. The yield of green plates of Ba<sub>2</sub>Th(S<sub>2</sub>)<sub>2</sub>S<sub>2</sub> was approximately 50 wt % based on Th, with the other product being ThOS.<sup>30</sup> Because ThOS and UOS are very stable<sup>30</sup> and both Th and U are oxyphilic, attack on the silica tubes, even if the tubes were carbon-coated, presumably led to these byproducts. Repeated attempts to optimize yields failed. Attempts to synthesize the Se or Te analogues failed; only BaQ and UQ<sub>2</sub> resulted. A few Ak/U/Se compounds have been synthesized at 1423 K,<sup>19</sup> which is beyond the temperature limit of our furnaces.

**Structure.** The isostructural compounds Ba<sub>2</sub>U(S<sub>2</sub>)<sub>2</sub>S<sub>2</sub> and Ba<sub>2</sub>Th(S<sub>2</sub>)<sub>2</sub>S<sub>2</sub> crystallize with two formula units in the tetragonal space group  $D_{4h}^{15}P4_2/nmc$  in a new structure type in cells with  $a = 5.4145(2)$  Å and  $c = 15.7322(5)$  Å [Ba<sub>2</sub>U(S<sub>2</sub>)<sub>2</sub>S<sub>2</sub>] and  $a = 5.4810(2)$  Å and  $c = 15.9860(6)$  Å [Ba<sub>2</sub>Th(S<sub>2</sub>)<sub>2</sub>S<sub>2</sub>]. The asymmetric unit contains one An atom (site symmetry  $\bar{4}m2$ ), one Ba atom ( $2mm$ ), and two S atoms [S1 ( $m$ ) and S2 ( $2mm$ )]. A general view of the structure approximately down [110] is presented in Figure 1. The structure consists of  ${}^2_\infty[\text{An}(\text{S}_2)_2(\text{S}_2)_4^-]$  layers (Figure 2) and Ba<sup>2+</sup> cations. The layers are stacked perpendicular to [001]. Each An center is coordinated to two S1–S1 pairs and four S2 atoms (Figure 3). Coordination may be described as pseudooctahedral if one considers the four S2 atoms and the centers of the two S1–S1 pairs. Metrical data are given in Table 2. The U–S distances are 2.8199(7) and 2.7337(2) Å for S1 and S2, respectively; the Th–S distances are 2.889(1) and 2.7759(3) Å. That the U–S distances are shorter than the corresponding Th–S distances is a manifestation of the actinide contraction. The U–S distances are typical for eight-coordinate U<sup>4+</sup> and comparable to those observed in K<sub>0.92</sub>U<sub>1.79</sub>S<sub>6</sub>, Rb<sub>0.85</sub>U<sub>1.74</sub>S<sub>6</sub>, and RbSbU<sub>2</sub>S<sub>8</sub> (Table 3). The Th–S distances are comparable to those of 2.844(2) and 2.968(1) Å for eight-coordinate Th<sup>4+</sup> in K<sub>10</sub>Th<sub>3</sub>(P<sub>2</sub>S<sub>7</sub>)<sub>4</sub>(PS<sub>4</sub>)<sub>2</sub><sup>47</sup> and may be compared to those for six-coordinate Th<sup>4+</sup> in KCuThS<sub>3</sub>, K<sub>2</sub>Cu<sub>2</sub>ThS<sub>4</sub>, and K<sub>3</sub>Cu<sub>3</sub>Th<sub>2</sub>S<sub>7</sub>.<sup>48</sup>

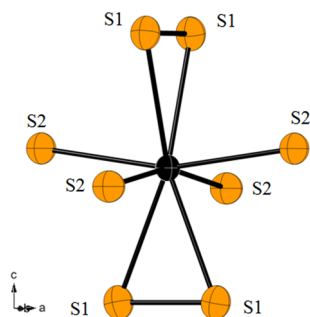
In Ba<sub>2</sub>U(S<sub>2</sub>)<sub>2</sub>S<sub>2</sub> and Ba<sub>2</sub>Th(S<sub>2</sub>)<sub>2</sub>S<sub>2</sub>, each Ba center (symmetry  $2mm$ ) is surrounded by six S1 and two S2 atoms, with Ba–S distances ranging between 3.2167(6) and 3.2978(7) Å and



**Figure 1.** Structure of  $\text{Ba}_2\text{An}(\text{S}_2)_2\text{S}_2$  (An = U, Th) approximately down the  $a$  axis.



**Figure 2.**  $2. \infty[\text{An}(\text{S}_2)_2(\text{S})_2]^{4-}$  layer in  $\text{Ba}_2\text{An}(\text{S}_2)_2\text{S}_2$  (An = U, Th).



**Figure 3.** Local coordination of the An (U, Th) center in  $\text{Ba}_2\text{An}(\text{S}_2)_2\text{S}_2$ . Symmetry  $4m2$  is imposed.

between 3.252(1) and 3.281(2) Å, respectively. These distances are comparable to the  $\text{BaS}_8$  interatomic distances of 3.123(9) and 3.609(7) Å in  $\text{BaUS}_3$ .<sup>17</sup>

The S1–S1 distances are 2.082(1) Å (U) and 2.098(2) Å (Th). Such distances are typical of a S–S single bond, for example, 2.055(2) Å in  $\text{S}_8$ ;<sup>49</sup> hence the formulation of the compounds as  $\text{Ba}_2\text{An}(\text{S}_2)_2\text{S}_2$ .

**Table 2.** Selected Interatomic Distances (Å) for  $\text{Ba}_2\text{U}(\text{S}_2)_2\text{S}_2$  and  $\text{Ba}_2\text{Th}(\text{S}_2)_2\text{S}_2$ <sup>a</sup>

|       | $\text{Ba}_2\text{U}(\text{S}_2)_2\text{S}_2$ | $\text{Ba}_2\text{Th}(\text{S}_2)_2\text{S}_2$ |
|-------|---|--|
| An–S1 | $2.8199(7) \times 4$                          | $2.889(1) \times 4$                            |
| An–S2 | $2.7337(2) \times 4$                          | $2.7759(3) \times 4$                           |
| S1–S1 | 2.082(1)                                      | 2.098(2)                                       |
| Ba–S1 | $3.2187(4) \times 4$                          | $3.2591(7) \times 4$                           |
| Ba–S1 | $3.2978(7) \times 2$                          | $3.281(1) \times 2$                            |
| Ba–S2 | $3.2167(6) \times 2$                          | $3.252(1) \times 2$                            |

<sup>a</sup>Atoms have the following site symmetries: An ( $\bar{4}m2$ ), Ba (2 mm.), S1 (m.), S2 (2 mm.).

Table 3 lists the An/S compounds in which S–S single bonds are present. The new structure adopted by  $\text{Ba}_2\text{U}(\text{S}_2)_2\text{S}_2$  and  $\text{Ba}_2\text{Th}(\text{S}_2)_2\text{S}_2$  comprises  $\text{Ba}^{2+}$  cations and  $2. \infty[\text{An}(\text{S}_2)_2(\text{S})_2]^{4-}$  layers (Figure 1). The  $\text{An}^{4+}$  cations in these layers are arranged linearly and are bridged by  $\text{S}^{2-}$  anions (Figure 2). This arrangement of  $\text{An}^{4+}$  cations is markedly different from those found in the layers in  $\text{US}_3$ ,<sup>5</sup>  $\text{K}_{0.92}\text{U}_{1.79}\text{S}_6$ ,<sup>50</sup> and  $\text{Rb}_{0.85}\text{U}_{1.74}\text{S}_6$ ,<sup>51</sup> these are based on the  $\text{ZrSe}_3$  structure.<sup>52</sup> In these structures, the layers are composed of double chains of  $\text{An}^{4+}$  cations (Figure 4). The layers in  $\text{RbSbU}_2\text{S}_8$ <sup>14</sup> are composed of similar  $\text{US}_8$  polyhedra. However, these share S atoms, with the Sb atoms situated in the centers of slightly distorted seesaws (Figure 4). In the structure of  $\text{Ta}_2\text{UO}(\text{S}_2)_3\text{Cl}_6$ ,<sup>15</sup> there are two different S–S bonds that are shared between U and Ta and a third S–S bond that is shared between Ta atoms. In the three-dimensional  $\text{An}_2\text{S}_5$  structures, the S–S bonds are shared by four An centers.

**Absorbance of  $\text{Ba}_2\text{Th}(\text{S}_2)_2\text{S}_2$ .** A fundamental absorption edge is clearly visible in the transmission measurements (Figure 5). Extrapolation to the absorption edge shows that  $\text{Ba}_2\text{Th}(\text{S}_2)_2\text{S}_2$  is a semiconductor. The direct and indirect band gaps derived from a total of 10 measurements on two different crystals are 2.46(5) and 2.42(6) eV, respectively; these are consistent with the light-green color of the crystals. A comparison of plots of absorbance versus energy ( $h\nu$ ; Figure 5, top) to plots of  $(ah\nu)^{1/2}$  (Figure 5, bottom left) and  $(ah\nu)^2$  (Figure 5, bottom right) versus  $h\nu$  indicates that the transition is direct. No polarization dependence was observed along the [001] direction of three different crystals when the polarization of visible light was scanned from 0 to 360°.

**Theoretical Results.** The computed total (DOS) and partial density of states (PDOS) are presented from –5 to +5 eV (the Fermi level is put at 0 eV) in Figures 6 and 7 for  $\text{Ba}_2\text{Th}(\text{S}_2)_2\text{S}_2$  and  $\text{Ba}_2\text{U}(\text{S}_2)_2\text{S}_2$ , respectively. For each atom, an orbital-resolved PDOS is plotted. Within the different magnetic orders allowed in the crystallographic cell,  $\text{Ba}_2\text{Th}(\text{S}_2)_2\text{S}_2$  is found to be nonmagnetic, whereas  $\text{Ba}_2\text{U}(\text{S}_2)_2\text{S}_2$  is found to be antiferromagnetic but almost degenerate with the ferromagnetic solution in terms of total energy. Therefore, for  $\text{Ba}_2\text{U}(\text{S}_2)_2\text{S}_2$  spin-polarized PDOS are presented.

From their total DOS, both compounds have sizable band gaps. In particular,  $\text{Ba}_2\text{Th}(\text{S}_2)_2\text{S}_2$  (Figure 6) has a band gap of 2.2 eV, with the top of the valence band being derived mainly from S1 p states, whereas the bottom of the conduction band is composed mainly of Th d states. The states from –5 to 0 eV originate from S1 and S2, whereas the conduction bands up to 5 eV come from Ba d, Th d, and Th f states. The PDOS of S2 and Th share similar features at –0.6 and –2.8 eV, which indicates a significant bonding between these two species. The PDOS of Ba is more difficult to analyze because it exhibits low

Table 3. Actinide Compounds Containing S–S Single Bonds

|  | structure         | An–S range (Å)      | S–S distance       | ref       |
|--|-------------------|---------------------|--------------------|-----------|
| Ba <sub>2</sub> U(S <sub>2</sub> ) <sub>2</sub> S <sub>2</sub>   | layered           | 2.7337(2)–2.8199(7) | 2.082(1)           | this work |
| Ba <sub>2</sub> Th(S <sub>2</sub> ) <sub>2</sub> S <sub>2</sub>  | layered           | 2.7759(3)–2.889(1)  | 2.098(2)           | this work |
| US <sub>3</sub>  | layered           | 2.753(2)–2.825(2)   | 2.086(4)           | 5         |
| K <sub>0.92</sub> U <sub>1.79</sub> S <sub>6</sub>               | layered           | 2.761(2)–2.846(2)   | 2.097(5)           | 50        |
| Rb <sub>0.85</sub> U <sub>1.74</sub> S <sub>6</sub>              | layered           | 2.775(3)–2.847(2)   | 2.106(5)           | 51        |
| RbSbU <sub>2</sub> S <sub>8</sub>                                | layered           | 2.752(3)–2.854(1)   | 2.096(5)           | 14        |
| Ta <sub>2</sub> UO(S <sub>2</sub> ) <sub>3</sub> Cl <sub>6</sub> | chains            | 2.819(3)–2.928(3)   | 2.081(4), 2.077(4) | 15        |
| U <sub>2</sub> S <sub>5</sub>                                    | three dimensional | 2.801–3.089         | 2.072              | 8         |
| Th <sub>2</sub> S <sub>5</sub>                                   | three dimensional | 2.861(4)–3.163(4)   | 2.117(7)           | 6–9       |

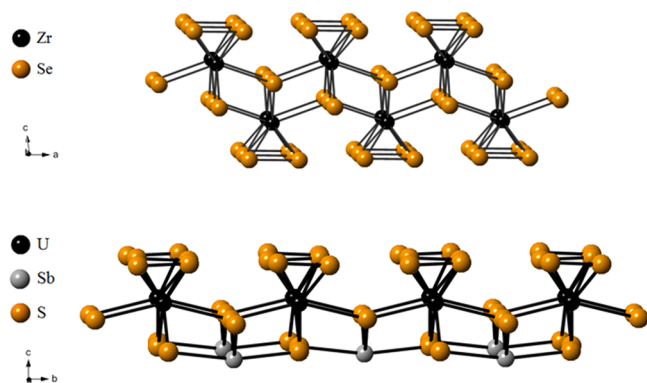


Figure 4. Arrangement of the cations in a layer of the ZrSe<sub>3</sub> structure<sup>52</sup> (top) and the RbSbU<sub>2</sub>S<sub>8</sub> structure<sup>14</sup> (bottom).

values for a wide range of energies below the Fermi level; these can be affected by the details of the numerical procedure.

In contrast, Ba<sub>2</sub>U(S<sub>2</sub>)<sub>2</sub>S<sub>2</sub> (Figure 7) has a band gap of 1.8 eV, with the top of the valence band made up of S2 p and U f

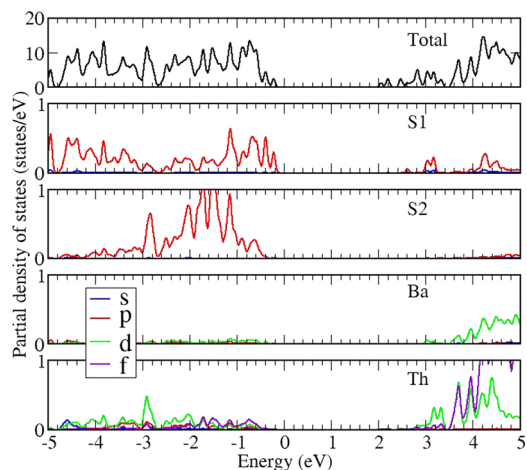


Figure 6. DOS (upper plot) and PDOS (lower plots) for Ba<sub>2</sub>Th(S<sub>2</sub>)<sub>2</sub>S<sub>2</sub>. For each atom, the PDOS is projected onto the relevant orbitals.

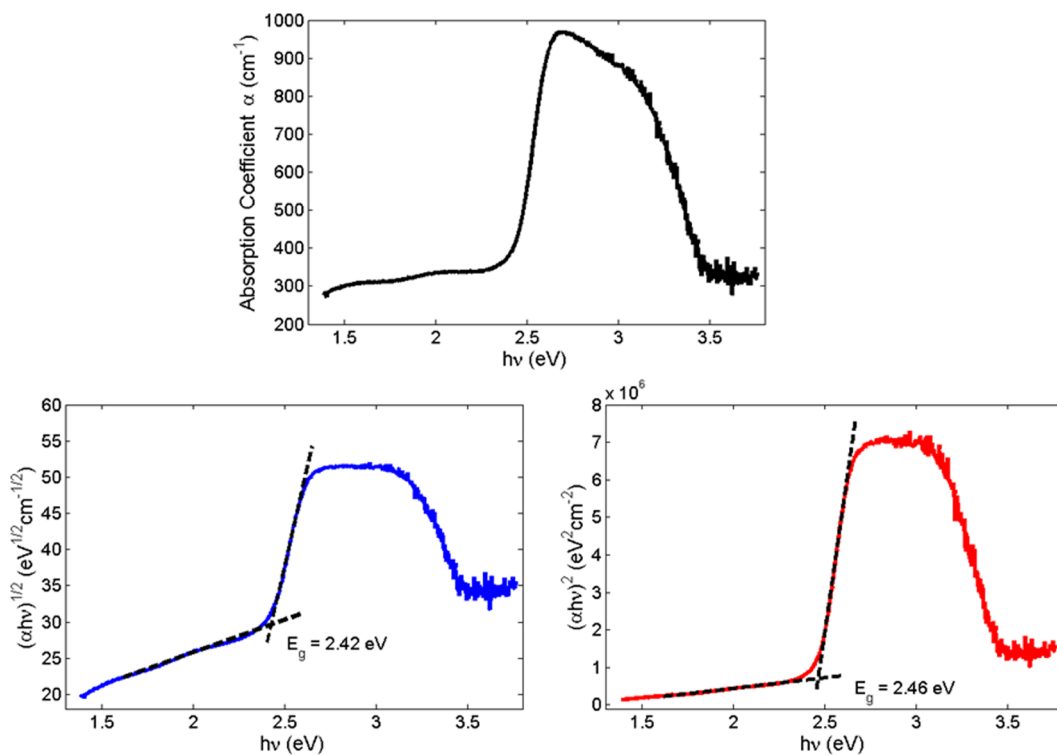
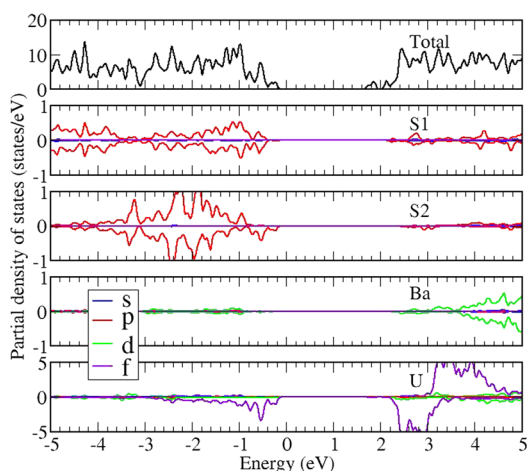


Figure 5. Absorbance of Ba<sub>2</sub>Th(S<sub>2</sub>)<sub>2</sub>S<sub>2</sub>. Top: absorbance ( $\alpha$ ) versus energy ( $h\nu$ ). Bottom left: spectrum calculated for an indirect band gap. Bottom right: spectrum calculated for a direct band gap.

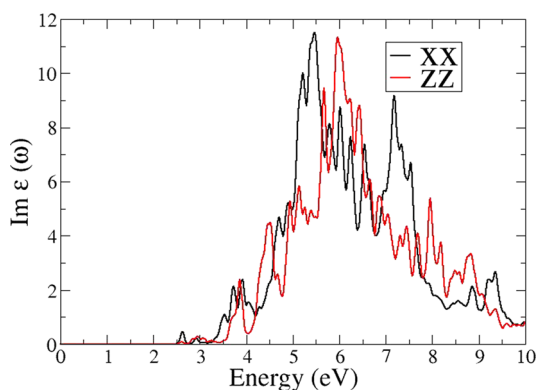




**Figure 7.** DOS (upper plot) and PDOS (lower plots) for  $\text{Ba}_2\text{U}(\text{S}_2)_2\text{S}_2$ . For each atom, the PDOS is projected onto the relevant orbitals.

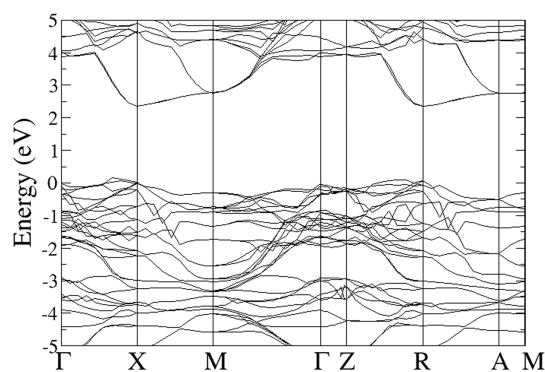
states. The bottom of the conduction band is more difficult to identify from the different PDOS. Although a small contribution appears to come from the unoccupied U f states, most correspond to the contribution to the total DOS from the interstitial region not covered by the spheres around each atom that were used to obtain the PDOS. As for  $\text{Ba}_2\text{Th}(\text{S}_2)_2\text{S}_2$ , the states from  $-5$  to  $0$  eV are derived from S1 and S2 states, but now for  $\text{Ba}_2\text{U}(\text{S}_2)_2\text{S}_2$ , the f-electron shell is partly filled and contributes to the total DOS for this range of energy. For the conduction bands up to  $5$  eV, all of the atoms are contributing. Also, the spin polarization of the U atom is clearly seen on the corresponding PDOS plot. It induces a spin polarization of the Ba and S2 atoms, whereas the S1 atoms seem to be less affected by the magnetic moment on the U atoms.

For a comparison with the optical measurements on  $\text{Ba}_2\text{Th}(\text{S}_2)_2\text{S}_2$ , we have calculated the imaginary part of the dielectric function for the two independent directions ( $\epsilon_{xx}$  and  $\epsilon_{zz}$ ) in the crystal (Figure 8). Both  $\epsilon_{xx}$  and  $\epsilon_{zz}$  show



**Figure 8.** Calculated imaginary part of the dielectric function of  $\text{Ba}_2\text{Th}(\text{S}_2)_2\text{S}_2$  for the two independent directions ( $\epsilon_{xx}$  and  $\epsilon_{zz}$ ) in the crystal.

many peaks that correspond to the different direct transitions allowed between the valence and conduction bands. Also, the observed threshold for transitions is positioned at about  $2.5$  eV, although the precision is limited because we have used a smearing technique to achieve integration over the Brillouin zone. The band structure (Figure 9) shows that the minimum direct transitions occur around the high-symmetry points X and



**Figure 9.** Calculated band structure for  $\text{Ba}_2\text{Th}(\text{S}_2)_2\text{S}_2$  along some high-symmetry directions.

R. In particular, the value of the corresponding direct band gap is  $2.4$  eV, which is in good agreement with the threshold observed in the imaginary part of the dielectric function. The gap of  $2.2$  eV observed in the total DOS corresponds to the indirect minimum band gap between the valence states along the X–M and Z–R directions and the conduction states at the X and R high-symmetry points. Because we have used the HSE functional, as opposed to the usual functionals, such as the local density approximation or GGA, or even the GGA+U approximation, to treat this strongly correlated system, our theoretical band gap is found to be in reasonable agreement<sup>38</sup> with the experimental value.

A quantitative understanding of the electron distribution in the crystal can be achieved with the use of Bader's analysis.<sup>53</sup> In order to carry out this analysis, the charge density in real space must be reconstructed properly from the projector-augmented wave basis set.<sup>54</sup> Applying Bader's analysis to our computed electron densities, we find that  $\text{Ba}_2\text{U}(\text{S}_2)_2\text{S}_2$  and  $\text{Ba}_2\text{Th}(\text{S}_2)_2\text{S}_2$  display very similar distributions of the valence electrons. These are  $6.7$ ,  $7.3$ ,  $8.4$ ,  $9.8$ , and  $11.8$  e<sup>−</sup> for S1, S2, Ba, Th, and U, respectively. Although some f electrons are present in  $\text{Ba}_2\text{U}(\text{S}_2)_2\text{S}_2$ , these have only a minor impact on the integrated charges per atom.

## CONCLUSIONS

Two new ternary actinide sulfides, namely,  $\text{Ba}_2\text{An}(\text{S}_2)_2\text{S}_2$  (An = U, Th), were synthesized at  $1273$  and  $1173$  K, respectively. Their structures were solved from single-crystal X-ray diffraction data. These isostructural compounds crystallize in the tetragonal space group  $P4_2/nmc$  and adopt a new structure type that comprises  $\text{Ba}^{2+}$  cations and  $[\text{An}(\text{S}_2)_2(\text{S}_2)_2]^{4-}$  layers. The  $\text{An}^{4+}$  cations in these layers are arranged linearly and are bridged by  $\text{S}^{2-}$  anions. This arrangement differs from those in other Ak/An/S compounds that contain S–S single bonds. Coordination about the An center, which has symmetry  $4m2$ , consists of two  $\text{S}_2^{2-}$  ions and four  $\text{S}^{2-}$  ions. This coordination may be described as pseudooctahedral if one considers the four S atoms and the centers of the two S–S pairs. The experimental optical direct band gap is  $2.46(5)$  eV for  $\text{Ba}_2\text{Th}(\text{S}_2)_2\text{S}_2$ ; this may be compared to the value of  $2.2$  eV from DFT calculations, performed with the HSE exchange–correlation potential, thus demonstrating the utility of applying this functional to 5f-electron systems. The calculated band gap of  $\text{Ba}_2\text{U}(\text{S}_2)_2\text{S}_2$  is  $1.8$  eV.

## ■ ASSOCIATED CONTENT

## ● Supporting Information

Crystallographic file in CIF format for  $\text{Ba}_2\text{U}(\text{S}_2)_2\text{S}_2$  and  $\text{Ba}_2\text{Th}(\text{S}_2)_2\text{S}_2$ . This material is available free of charge via the Internet at <http://pubs.acs.org>.

## ■ AUTHOR INFORMATION

## Corresponding Author

\*E-mail: [ibers@chem.northwestern.edu](mailto:ibers@chem.northwestern.edu).

## Notes

The authors declare no competing financial interest.

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## ■ REFERENCES

- (1) Manos, E.; Kanatzidis, M. G.; Ibers, J. A. In *The Chemistry of the Actinide and Transactinide Elements*, 4th ed.; Morss, L. R., Edelstein, N. M., Fuger, J., Eds.; Springer: Dordrecht, The Netherlands, 2010; Vol. 6, pp 4005–4078.
- (2) Bugaris, D. E.; Ibers, J. A. *Dalton Trans.* **2010**, 39, 5949–5964.
- (3) Sunshine, S. A.; Kang, D.; Ibers, J. A. *J. Am. Chem. Soc.* **1987**, 109, 6202–6204.
- (4) Narducci, A. A.; Ibers, J. A. *Inorg. Chem.* **1998**, 37, 3798–3801.
- (5) Kwak, J.-e.; Gray, D. L.; Yun, H.; Ibers, J. A. *Acta Crystallogr., Sect. E: Struct. Rep. Online* **2006**, 62, i86–i87.
- (6) Graham, J.; McTaggart, F. K. *Aust. J. Chem.* **1960**, 13, 67–73.
- (7) Kohlmann, H.; Beck, H. P. Z. *Kristallogr.* **1999**, 214, 341–345.
- (8) Noël, H. *J. Inorg. Nucl. Chem.* **1980**, 42, 1715–1717.
- (9) Noël, H.; Potel, M. *Acta Crystallogr., Sect. B: Struct. Crystallogr. Cryst. Chem.* **1982**, 38, 2444–2445.
- (10) Cody, J. A.; Ibers, J. A. *Inorg. Chem.* **1995**, 34, 3165–3172.
- (11) Sutorik, A. C.; Kanatzidis, M. G. *J. Am. Chem. Soc.* **1991**, 113, 7754–7755.
- (12) Cody, J. A.; Ibers, J. A. *Inorg. Chem.* **1996**, 35, 3836–3838.
- (13) Choi, K.-S.; Patschke, R.; Billinge, S. J. L.; Waner, M. J.; Dantus, M.; Kanatzidis, M. G. *J. Am. Chem. Soc.* **1998**, 120, 10706–10714.
- (14) Choi, K.-S.; Kanatzidis, M. G. *Chem. Mater.* **1999**, 11, 2613–2618.
- (15) Wells, D. M.; Chan, G. H.; Ellis, D. E.; Ibers, J. A. *J. Solid State Chem.* **2010**, 183, 285–290.
- (16) Brochu, R.; Padiou, J.; Grandjean, D. C. R. *Seances Acad. Sci., Ser. C* **1970**, 271, 642–643.
- (17) Lelieveld, R.; Ijdo, D. J. W. *Acta Crystallogr., Sect. B: Struct. Crystallogr. Cryst. Chem.* **1980**, 36, 2223–2226.
- (18) Brochu, R.; Padiou, J.; Prigent, J. C. R. *Acad. Sci. Paris* **1970**, 270, 809–810.
- (19) Brochu, R.; Padiou, J.; Prigent, J. C. R. *Seances Acad. Sci., Ser. C* **1972**, 274, 959–961.
- (20) Yao, J.; Ibers, J. A. *Z. Anorg. Allg. Chem.* **2008**, 634, 1645–1647.
- (21) Bugaris, D. E.; Ibers, J. A. *Inorg. Chem.* **2012**, 51, 661–666.
- (22) Bugaris, D. E.; Ibers, J. A. *J. Solid State Chem.* **2008**, 181, 3189–3193.
- (23) Haneveld, A. J. K.; Jellinek, F. J. *Less-Common Met.* **1969**, 18, 123–129.
- (24) APEX2, version 2009.5-1, and SAINT, *Data Collection and Processing Software*, version 7.34a; Bruker Analytical X-ray Instruments, Inc.: Madison, WI, USA, 2009.
- (25) SMART, *Data Collection*, version 5.054, and SAINT-Plus, *Data Processing Software for the SMART System* version 6.45a; Bruker Analytical X-ray Instruments, Inc.: Madison, WI, 2003.
- (26) Sheldrick, G. M. *Acta Crystallogr., Sect. A: Found. Crystallogr.* **2008**, 64, 112–122.
- (27) Gelato, L. M.; Parthé, E. *J. Appl. Crystallogr.* **1987**, 20, 139–143.
- (28) Spek, A. L. *PLATON, A Multipurpose Crystallographic Tool*; Utrecht University: Utrecht, The Netherlands, 2008.
- (29) Oh, G. N.; Ringe, E.; Van Duyne, R. P.; Ibers, J. A. *J. Solid State Chem.* **2012**, 185, 124–129.
- (30) Koscielski, L. A.; Ringe, E.; Van Duyne, R. P.; Ellis, D. E.; Ibers, J. A. *Inorg. Chem.* **2012**, 51, 8112–8118.
- (31) Amoretti, G.; Blaise, A.; Caciuffo, R.; Fournier, J. M.; Larroque, J.; Osborn, R. *J. Phys.: Condens. Matter* **1989**, 1, 5711–5720.
- (32) Ballestracci, R.; Bertaut, E. F.; Pauthenet, R. *J. Phys. Chem. Solids* **1963**, 24, 487–491.
- (33) Hohenberg, P.; Kohn, W. *Phys. Rev.* **1964**, 136, 864–871.
- (34) Heyd, J.; Scuseria, G. E.; Ernzerhof, M. *J. Phys. Chem.* **2003**, 118, 8207–8215.
- (35) Paier, J.; Marsman, M.; Hummer, K.; Kresse, G.; Gerber, I. C.; Angyan, J. G. *J. Chem. Phys.* **2006**, 125, 249901–1–2.
- (36) Heyd, J.; Scuseria, G. E.; Ernzerhof, M. *J. Chem. Phys.* **2006**, 124, 219906–1.
- (37) Perdew, J. P.; Burke, K.; Ernzerhof, M. *Phys. Rev. Lett.* **1996**, 77, 3865–3868.
- (38) Henderson, T. M.; Paier, J.; Scuseria, G. E. *Phys. Status Solidi B* **2011**, 248, 767–774.
- (39) Prodan, I. D.; Scuseria, G. E.; Martin, R. L. *Phys. Rev. B* **2007**, 76, 033101-1–033101-4.
- (40) Hay, P. J.; Martin, R. L.; Uddin, J.; Scuseria, G. E. *J. Chem. Phys.* **2006**, 125, 034712-1–034712-8.
- (41) Da Silva, J. L. F.; Ganduglia-Pirovano, M. V.; Sauer, J.; Bayer, V.; Kresse, G. *Phys. Rev. B* **2007**, 75, 045121-1–045121-10.
- (42) Ganduglia-Pirovano, M. V.; Da Silva, J. L. F.; Sauer, J. *Phys. Rev. Lett.* **2009**, 102, 026101-1–026101-4.
- (43) Kresse, G.; Furthmüller, J. *Comput. Mater. Sci.* **1996**, 6, 15–50.
- (44) Kresse, G.; Joubert, D. *Phys. Rev. B* **1999**, 59, 1758–1775.
- (45) Blöchl, P. E. *Phys. Rev. B* **1994**, 50, 17953–17979.
- (46) Gajdos, M.; Hummer, K.; Kresse, G.; Furthmüller, J.; Bechstedt, F. *Phys. Rev. B* **2006**, 73, 045112–1–9.
- (47) Hess, R. F.; Abney, K. D.; Berris, J. L.; Hochheimer, H. D.; Dorhout, P. K. *Inorg. Chem.* **2001**, 40, 2851–2859.
- (48) Selby, H. D.; Chan, B. C.; Hess, R. F.; Abney, K. D.; Dorhout, P. K. *Inorg. Chem.* **2005**, 44, 6463–6469.
- (49) Rettig, S. J.; Trotter, J. *Acta Crystallogr., Sect. C: Cryst. Struct. Commun.* **1987**, 43, 2260–2262.
- (50) Mizoguchi, H.; Gray, D.; Huang, F. Q.; Ibers, J. A. *Inorg. Chem.* **2006**, 45, 3307–3311.
- (51) Bugaris, D. E.; Wells, D. M.; Yao, J.; Skanthakumar, S.; Haire, R. G.; Soderholm, L.; Ibers, J. A. *Inorg. Chem.* **2010**, 49, 8381–8388.
- (52) Krönert, W.; Plieth, K. *Z. Anorg. Allg. Chem.* **1965**, 336, 207–218.
- (53) Bader, R. F. W. *Atoms in molecules: a quantum theory*; International Series of Monographs on Chemistry 22; Oxford University Press, Inc.: New York, 1990.
- (54) Aubert, E.; Lebègue, S.; Marsman, M.; Thu Bui, T. T.; Jelsch, C.; Dahaoui, S.; Espinosa, E.; Angyán, J. G. *J. Phys. Chem.* **2011**, A115, 14484–14494.